



# Assessing the influence of humic acids on the weathering of galena and its environmental implications

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## ABSTRACT

Galena weathering often occurs in nature and releases metal ions during the process. Humic acid (HA), a critical particle of natural organic matter, binds metal ions, thus affecting metal transfer and transformation. In this work, an electrochemical method combined with spectroscopic techniques was adopted to investigate the interfacial processes involved in galena weathering under acidic and alkaline conditions, as well as in the presence of HA. The results show that the initial step of galena weathering involved the transformation  $Pb^{2+}$  and  $S^0$ , regardless of whether the solution was acidic or alkaline. Under acidic conditions,  $S^0$  and  $Pb^{2+}$  further transform into anglesite, and HA adsorbs on the galena surface, inhibiting the transformation of sulfur. HA and Pb (II) ions form bridging complexes. Under alkaline conditions without HA, the sulfur produced undergoes no transformation, whereas  $Pb^{2+}$  will transform into PbO. The presence of HA changes the galena weathering mechanism via ionization effect, and  $Pb^{2+}$  is ultimately converted into anglesite. Higher acidity in acidic conditions or higher alkalinity in alkaline conditions causes galena corrosion when the electrolyte does not contain HA. Conversely, higher pH always accelerates galena corrosion when the electrolyte contains HA, whether the electrolyte is acidic or alkaline. At the same acidity/alkalinity, increasing the concentration of HA inhibits galena weathering. Galena will release  $134.7 \text{ g m}^{-2} \text{ y}^{-1} Pb^{2+}$  to solution at pH 2.5, and the amount decreases to  $28.09 \text{ g m}^{-2} \text{ y}^{-1}$  in the presence of 1000 mg/L HA. This study provides an in situ electrochemical method for the assessment of galena weathering.

## 1. Introduction

Galena, one of the most common lead-bearing sulfide minerals, can easily weather when exposed to oxidizing conditions and form more stable secondary lead species (Benvenuti et al., 2000; Keim and Markl, 2015). In natural geologic environments, anglesite and cerussite are most often transformed during galena weathering (Lara et al., 2011; Szczerba and Sawlowicz, 2009). During the galena weathering process, heavy metals, such as Pb, As, and Cr, are released and cause ecological risk to the soil-plant system and the sediment-water column (Azhari et al., 2017). In a laboratory setting, Hsieh and Huang (1989) were the first to systematically study the dissolution of PbS(s) in dilute aqueous solutions. The results show that the dissolution reaction is pH dependent, enhanced by dissolved oxygen and slightly affected by photo-irradiation. The overall dissolution processes is controlled by surface reaction. Based on this study, Fornasiero et al. (1994) used Fourier transform infrared spectroscopy and X-ray photoelectron spectroscopy to investigate the surface oxidation of galena subject to various conditioning environments. They proposed a mechanism of oxidation

involving the dissolution of lead and sulfide ions, the formation of lead sulfoxy species and lead carbonate, and their subsequent adsorption or precipitation on the galena surface. Similarly, they determined the oxidation reaction is pH dependent, and the oxidative dissolution of PbS(s) occurs primarily by the reaction (1), as Hsieh and Huang (1989) reported. In recent years, isotope technique has been a useful tool for the investigation of galena oxidation. Heidel and Tichomirowa (2011) investigated galena oxidation mechanisms via oxygen and sulfur isotopes. The oxygen isotope composition of sulphate produced from galena oxidation could be determined for the first time. Sulfur isotopes of sulphate showed an enrichment of  $(32) S^0$  in sulphate (relative to galena), which increases with increasing pH. Moreover, its enrichment processes may be associated with the formation of anglesite.



Under natural conditions, many studies (Acero et al., 2007; Gerson and O'Dea, 2003; Keim and Markl, 2015; Mikhlín et al., 2006; Ryan et al., 2001) have revealed the mechanism of galena weathering via

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characteristics of galena weathering. Polarization curves allowed for immediate weathering rate information. EIS measurements were carried out at different AC frequencies, and analysis of the system response revealed information about the interface, its structure and reactions taking place there. The polarization curves were obtained by changing the electrode potential automatically from  $-250$  to  $+250$  mV (vs. open current potential, OCP) at a scan rate of  $10 \text{ mV s}^{-1}$ . The EIS tests were performed using the OCP in the frequency range of  $0.001$ – $10,000$  Hz, with a peak-to-peak amplitude of  $10$  mV. ZSimpWin 3.20 (2004) software was used to fit the impedance data. To ensure reproducibility, identical experiments were repeated at least three times (the random errors of the results of all three identical experimental were within tolerance), and all results reported in this paper were averaged. The experiments were conducted at  $25 \pm 1$  °C.

Prior to the polarization and EIS tests, OCP tests were performed. During the OCP experiments, the electrode potential increased for  $5$  min and reached a quasi-steady state, where the steady-state is defined here as a change of less than  $2$  mV per  $5$  min. Then, the electrode was stabilized for  $400$  s, and the potential was recorded as the OC potential. When the duplicate/triplicate experiments were performed, if the potential was not within  $\pm 5$  mV of the first experiment when the quasi-steady state was reached, the OCP test ceased. Then, a new replicate was performed to obtain the same OC potential as the first test when stabilized for approximately  $400$  s. In this work, the standard deviations of the experimental results no more than  $3\%$  were reported.

#### 2.4. Surface characterization measurements

The surface morphologies of corroded samples were investigated by

Fourier transform infrared (FTIR) spectroscopy and Raman spectroscopy. Seven  $0.5 \text{ cm} \times 0.5 \text{ cm} \times 0.01 \text{ cm}$  galena samples were used for FTIR analysis. The seven galena samples were immersed in seven different solutions: pH 4.5 without HA; pH 4.5 with  $10$  and  $100$  mg/L HA, respectively; pH 8.5 without HA; pH 8.5 with  $10$  and  $100$  mg/L HA, respectively, and the seventh solution was pH 8.5 with  $100$  mg/L HA plus  $1 \text{ mg/L Pb}^{2+}$ . All the samples were weathered for  $10$  weeks and then characterized using FTIR. FTIR spectra were collected on a Bruker Vertex 70 Fourier transform infrared spectrometer with an Attenuated Total Reflectance (ATR) accessory. The scanning range was  $600$ – $4000 \text{ cm}^{-1}$  with a resolution of  $4 \text{ cm}^{-1}$  and  $16$  scans.

Four  $1.0 \text{ cm} \times 1.0 \text{ cm} \times 0.2 \text{ cm}$  galena samples were used for surface analysis. The four galena samples were immersed in four different solutions: (1) pH 4.5 with  $100$  mg/L HA, (2) pH 4.5 without HA, (3) pH 8.5 with  $100$  mg/L HA and (4) pH 8.5 without HA. The four pristine samples and the samples weathered for  $10$  weeks were characterized using Raman spectroscopy (British Renishaw inVia Reflex type microscopic confocal laser Raman spectrometer) at a wavelength of  $514 \text{ nm}$  and a collection time of  $10$  s. The laser power was  $50 \text{ mW}$  to avoid the destruction of the galena samples.

### 3. Results and discussion

#### 3.1. Polarization curves study

Fig. 2 shows the potentiodynamic curves of the galena electrode at different concentrations of HA solutions. The potentiodynamic curves show that the galena electrode had a similar potentiodynamic profile, suggesting the same electrochemical interaction mechanism, in the

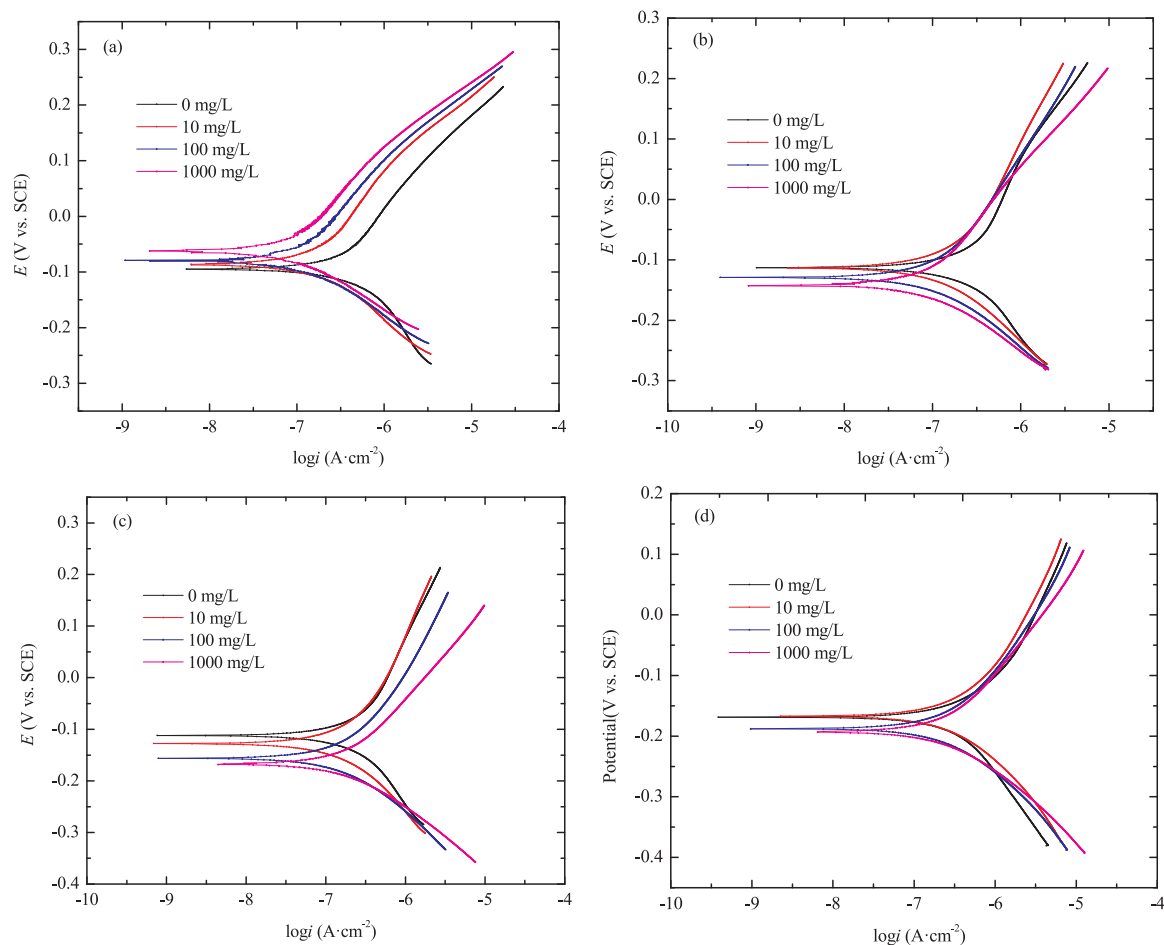


Fig. 2. Potentiodynamic curves of galena in different concentrations of HA solutions (25 °C, a: pH 2.5; b: pH 4.5; c: pH 8.5; d: pH 10.5).

**Table 1**  
Electrochemical parameters of the galena electrode under different concentrations and acidities of HA solutions.

pH	C <sub>HA</sub> (mg L <sup>-1</sup> )	E <sub>corr</sub> (mV)	i <sub>corr</sub> (μA cm <sup>-2</sup> )	b <sub>c</sub> (mV)	b <sub>a</sub> (mV)	R <sub>p</sub> (KΩ cm <sup>2</sup> )
2.5	0	-93.20	0.398	168.83	213.62	103.0
	10	-84.72	0.155	143.42	206.92	237.6
	100	-75.46	0.105	100.32	192.57	273.1
	1000	-59.34	0.083	97.46	169.90	324.4
4.5	0	-112.51	0.303	192.84	294.63	167.2
	10	-115.40	0.216	172.46	272.59	212.6
	100	-128.89	0.143	132.84	236.42	258.6
	1000	-140.18	0.107	102.16	177.36	263.4
8.5	0	-115.79	0.469	300.48	457.58	168.1
	10	-127.86	0.318	234.77	397.59	201.8
	100	-156.05	0.246	198.14	372.05	228.5
	1000	-163.20	0.137	134.60	202.91	256.8
10.5	0	-162.60	0.815	325.66	282.64	80.7
	10	-167.37	0.540	186.23	258.74	87.2
	100	-187.64	0.382	168.30	213.37	107.1
	10.5	-181.12	0.243	120.22	173.32	127.0

E<sub>corr</sub>: corrosion potential; i<sub>corr</sub>: corrosion current density; b<sub>a</sub>: anode Tafel slope; b<sub>c</sub>: cathode Tafel slope; R<sub>p</sub>: polarization resistance.

presence or absence of HA electrolyte.

The galena polarization parameters, corrosion potential (E<sub>corr</sub>) and corrosion current density (i<sub>corr</sub>) were dependent on Tafel extrapolation (Bard and Faulkner, 2001), and polarization resistance R<sub>p</sub> was also obtained from the Stern-Geary equation (Eq. 2) (Stern and Geary, 1957).

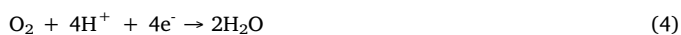
$$R_p = b_a b_c / [2.3 i_{corr} (b_a + b_c)] \quad (2)$$

For these electrochemical parameters, the parameter i<sub>corr</sub> reflects the corrosion rate of the material. Specifically, the higher the density of current is, the faster the corrosion rate will be, while E<sub>corr</sub> is related to the susceptibility of the material to corrosion. A negative value of E<sub>corr</sub> means the material is difficult to corrode. A bigger Tafel slope indicates greater interaction resistance, i.e., unfavorable conditions for electrode interaction. The parameter R<sub>p</sub> reflects the interaction resistance. A bigger R<sub>p</sub> value means the electrode process changes more difficult. The detailed polarization results are listed in Table 1. The polarization results of galena samples exposed to different concentrations of HA revealed that at the same pH, a higher HA concentration resulted in a smaller corrosion density (i<sub>corr</sub>). This suggests that HA inhibits the electrochemical corrosion of galena, which comes from the increasing resistance of the polarization of the galena samples.

When the pH of the HA electrolyte solution was 2.5 and the concentration of HA increased from 0 mg/L to 10 mg/L, the galena corrosion density (i<sub>corr</sub>) decreased from 0.398 to 0.155 μA cm<sup>-2</sup>. Then, the corrosion density decreased to 0.105 μA cm<sup>-2</sup> at a HA concentration of 100 mg/L, and finally decreased to 0.083 μA cm<sup>-2</sup> at a HA concentration of 1000 mg/L. Therefore, increasing the concentration of HA inhibits galena weathering, and the inhibiting efficiencies (η) were 61.06%, 73.62% and 79.15%, respectively. In material science, η is defined by Eq. (3) (Solmaz et al., 2008):

$$\eta = (i_{corr} - i_{corr}^0) / i_{corr}^0 \quad (3)$$

where i<sub>corr</sub><sup>0</sup> and i<sub>corr</sub> are the corrosion current densities without and with HA, respectively. The reaction involved in the natural weathering of galena is undoubtedly an electrochemical process, and the anodic reaction is expressed as reaction (1) (Paul et al., 1978). O<sub>2</sub> is reduced on the cathode (reaction (4)) when the electrolyte is acidic.



As mentioned previously, HA has two polar functional groups, a phenolic group and a carboxylic group. HA can ionize and release H<sup>+</sup> and -COO<sup>-</sup> ions. A higher concentration of HA releases higher concentrations of H<sup>+</sup> and -COO<sup>-</sup> ions. At the cathode, the higher concentration of H<sup>+</sup> is favorable for O<sub>2</sub> reduction, and this is the reason why the cathodic Tafel slope (b<sub>c</sub>) decreased. At the anode, the higher

concentrations of -COO<sup>-</sup> ions favor the oxidation reaction of the complexation of -COO<sup>-</sup> and Pb<sup>2+</sup>. This complexation causes the anodic Tafel slope (b<sub>a</sub>) to decrease. It seems that the higher concentration of HA promotes galena electrochemical corrosion, but other facts cannot be ignored. When the complexation of -COO<sup>-</sup> and Pb<sup>2+</sup> is enhanced, more S<sup>0</sup> is also produced. Thus, when HA is present in higher concentrations, more HA and S<sup>0</sup> are absorbed on the electrode surface and passivate the galena electrode. This passivation results in an increase in the galena electrode polarization resistance (R<sub>p</sub>), which causes the galena corrosion current density to decrease, and the galena weathering is weakened. When the pH of the HA electrolyte was 4.5, i<sub>corr</sub> decreased from 0.303 to 0.216 μA cm<sup>-2</sup> and to 0.143 μA cm<sup>-2</sup> with increasing HA concentration from 0 mg/L to 10 mg/L and to 100 mg/L, respectively. Finally, the corrosion density reached 0.107 μA cm<sup>-2</sup> when HA increased to 1000 mg/L. The inhibiting efficiencies were 28.71%, 52.81% and 64.69%, respectively. The cause and explanation for this are similar to that of the pH-dependence discussed above.

When the pH of the HA electrolyte was 8.5 the galena i<sub>corr</sub> decreased from 0.469 to 0.318 μA cm<sup>-2</sup> and to 0.246 μA cm<sup>-2</sup> with increasing HA concentration from 0 mg/L to 10 mg/L and to 100 mg/L, respectively. Finally, the corrosion density reached 0.137 μA cm<sup>-2</sup> when the HA concentration was 1000 mg/L. The inhibiting efficiencies were 32.19%, 47.55% and 70.79%, respectively. When the electrolyte pH was 10.5, the galena i<sub>corr</sub> decreased from 0.815 to 0.540 μA cm<sup>-2</sup> and from 0.382 μA cm<sup>-2</sup> to 0.243 μA cm<sup>-2</sup> when the HA concentration was 1000 mg/L. The inhibiting efficiencies were 33.74%, 53.13% and 70.18%, respectively. A detailed explanation for the above phenomena is presented as follows: at the anode, the galena was oxidized according to reaction (1). Similar to that of an acidic electrolyte, a higher concentration of HA provided greater HA ionization and released more H<sup>+</sup> and -COO<sup>-</sup> ions. Higher concentrations of -COO<sup>-</sup> ions favor the anode reaction, which is the complexation of -COO<sup>-</sup> and Pb<sup>2+</sup> ions. This complexation is why the anodic Tafel slope (b<sub>a</sub>) decreased. At the cathode, higher concentrations of H<sup>+</sup> ions favor the cathode reaction (reaction (4)), causing the cathodic Tafel slope (b<sub>c</sub>) to decrease. Corrosion potential (E<sub>corr</sub>) is another important parameter to evaluate the electrochemical corrosion of a material. Generally, a more positive corrosion potential suggests that the material has better corrosion resistance. When the electrolyte pH was 2.5, as mentioned above, increasing the concentration of HA inhibited galena corrosion. As the galena corrosion current density decreased, the corrosion potential became more positive. These data suggested galena had a better corrosion resistance when increasing the concentration of HA. However, when the electrolyte pH was 4.5, 8.5 or 10.5, the galena corrosion current density decreased, and the corrosion potential changed more negatively. The corrosion potential results suggest that the corrosion



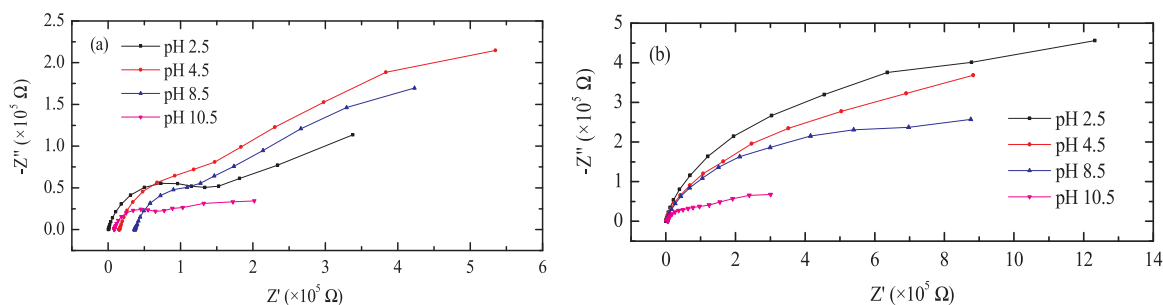
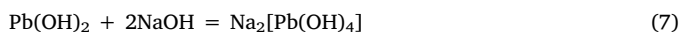


Fig. 3. Nyquist plots for galena in 0 mg/L (a) and 100 mg/L (b) HA solutions with different pH.

resistance of the galena samples decreased with increasing concentrations of HA. This is because HA has a relatively open structure and contains numerous functional groups that in aqueous solutions, especially in weakly acidic or alkaline solutions, become predominantly negatively charged (such as  $-\text{COO}^-$  ions) because of proton dissociation (Ren et al., 2015).

Fig. S1 shows the potentiodynamic curves of the galena electrode in HA solutions at different pH values. When the electrolyte contained HA, HA did not change the profiles of the potentiodynamic curves, suggesting the same electrochemical interaction mechanism. In contrast, the potentiodynamic curve shifted to the right and became more negative in the x and y directions when the pH increased, suggesting an increase in corrosion current density and a shift to more negative corrosion potential.

When the electrolyte did not contain HA (Fig. S1 (a)), the polarization parameters in Table 1 show that the galena  $i_{\text{corr}}$  increased from 0.303 to 0.398  $\mu\text{A cm}^{-2}$ , as the pH decreased from 4.5 to 2.5. This indicates that higher acidity accelerates galena corrosion. The promotion efficiency was 23.87%, and the polarization resistance reduced from 167.2 to 103.0  $\text{k}\Omega \text{cm}^2$ . The causes for this phenomenon are presented as follows: (1) in acidic electrolyte, galena acted as an anode and was oxidized according to reaction (1); (2)  $\text{S}^\ominus$  was produced and absorbed on the galena electrode surface, and (3)  $\text{O}_2$  was reduced at the cathode according to reaction (4). The higher acidity favors  $\text{S}^\ominus$  dissolution and is beneficial for the reduction of  $\text{O}_2$ . Thus, this acidity causes the anodic and cathodic Tafel slopes to decrease and ultimately results in a decrease in the polarization resistance. In alkaline electrolyte, the galena  $i_{\text{corr}}$  increased from 0.469 to 0.815  $\mu\text{A cm}^{-2}$ , as pH increased from 8.5 to 10.5, and the promotion efficiency was 73.77% when the polarization resistance decreased from 168.1 to 80.7  $\text{k}\Omega \text{cm}^2$ . In contrast to the acidic electrolyte, the decrease in the polarization resistance with increasing alkalinity was because of the decrease in the anodic Tafel slope, despite the increase in the cathodic Tafel slope. A detailed explanation is as follows: in the alkaline electrolyte,  $\text{O}_2$  was reduced via reaction (5) at the cathode. The higher concentration of  $\text{OH}^-$  ions was unfavorable for reaction (5), resulting in a larger cathodic Tafel slope. However, the higher concentrations of  $\text{OH}^-$  ions favor the galena anodic oxidation as in reactions (6) and (7), which caused the anodic Tafel slope to decrease.



When the electrolyte contained 10 mg/L, 100 mg/L and 1000 mg/L HA (Fig. S1 (b-d)), the galena polarization results in Table 1 show the galena  $i_{\text{corr}}$  increased as the HA solution pH increased. This indicates that stronger alkalinity promotes galena corrosion. Meanwhile, galena  $E_{\text{corr}}$  changed more negatively, which implies galena has excellent erosion-resistance. This erosion-resistance is because galena polarization resistance decreased as the solution pH increased.

The presence of HA had no effect on the galena electrochemical

interaction mechanism. Under acidic conditions,  $\text{H}^+$  ions at the cathode participate in  $\text{O}_2$  reduction via reaction (4). Thus, reducing the acidity (increasing pH from 2.5 to 4.5) was unfavorable for the reduction of  $\text{O}_2$ , and this was why the cathodic Tafel slope ( $b_c$ ) increased. At the anode, galena was oxidized via reaction (1), and when the pH of the HA solution increased from 2.5 to 4.5, the HA ionization changed more strongly, releasing more  $\text{H}^+$  and  $-\text{COO}^-$  ions. This release provoked the complexation of  $-\text{COO}^-$  ions and  $\text{Pb}^{2+}$  ions, resulting in a decrease in polarization resistance. However, the increase in passivated  $\text{S}^\ominus$  inhibited the galena anodic oxidation. This, in turn, resulted in an increase in the anodic Tafel slope ( $b_a$ ). Under alkaline conditions,  $\text{O}_2$  at the cathode was reduced via reaction (5). It seemed that the cathodic Tafel slope ( $b_c$ ) should increase following the pH increase from 8.5 to 10.5 for the electrolyte without HA, as higher concentrations of  $\text{OH}^-$  ions were unfavorable for reaction (5). However, due to the higher concentrations of  $\text{OH}^-$  ions, HA ionizes in the presence of the electrolyte, consuming  $\text{OH}^-$  ions and accelerating reaction (5), resulting in a decrease of the cathodic Tafel slope ( $b_c$ ). At the anode, galena was oxidized via reaction (1), and when the pH of the HA solution increased from 8.5 to 10.5, the HA ionization increased, releasing more  $\text{H}^+$  and  $-\text{COO}^-$  ions. This release provoked the complexation of  $-\text{COO}^-$  ions and  $\text{Pb}^{2+}$  ions, resulting in an anodic Tafel slope ( $b_a$ ) decrease. As mentioned above, under alkaline conditions,  $\text{Pb}^{2+}$  from galena can transform into  $\text{Pb}[(\text{OH})_4]^{2-}$  via reactions (6) and (7), as well as promote anodic oxidation.

### 3.2. EIS study

EIS has become a ubiquitous technique for the analysis of fundamental diffusion and Faradaic reactions, such as corrosion, adsorption, and chemical reactions coupled with Faradaic processes at electrodes (Lasia, 2002). In this study, EIS studies were conducted to confirm the electrochemical corrosion processes of galena. Fig. 3(a) and Fig. 3(b) present the Nyquist plots for galena in 0 mg/L and 100 mg/L HA solutions, respectively, at different pH values. Regardless of the presence or concentration of HA, the Nyquist plots exhibited two capacitive loops. The loop at high frequency was attributed to the charge transfer resistance ( $R_c$ ) corresponding to the resistance between the galena and the outer Helmholtz plane. The loop at low frequency, which is slightly distorted, is related to the combination of the pseudo-capacitance impedance (because of the passive layer) and resistance  $R_f$ . The deviation from an ideal semicircle was attributed to the frequency dispersion and the inhomogeneity of the passive layer surface. The electrochemical equivalent circuit (EEC) shown in Fig. S2 was used to model the galena/electrolyte interface. In this EEC,  $R_s$  is the ohmic resistance of the solution;  $R_c$  is the charge transfer resistance;  $R_f$  is the passive film resistance; and  $\text{CPE}_i$  and  $\text{CPE}_f$  represent the constant phase element used to replace the charge transfer capacitance at the double layer ( $C_c$ ) and the passive film capacitance ( $C_f$ ), respectively. The impedance of the CPE is given by Macdonald (1985) as:

$$Z_{\text{CPE}} = \frac{1}{Y_0(j\omega)^n} \quad (8)$$

**Table 2**  
Equivalent circuit model parameters for galena in electrolyte without and with 100 mg/L HA.

pH	$C_{HA}$ (mg L <sup>-1</sup> )	$CPE_{\text{e}}, Y_0$ (S cm <sup>-2</sup> s <sup>n</sup> )	$n$	$R_t$ ( $\Omega$ cm <sup>2</sup> )	$CPE_{\text{f}}, Y_0$ (S cm <sup>-2</sup> s <sup>n</sup> )	$n$	$R_f$ ( $\Omega$ cm <sup>2</sup> )
2.5	0	$6.906 \times 10^{-5}$	0.8204	$3.956 \times 10^4$	$1.992 \times 10^{-3}$	1	$4.647 \times 10^4$
	100	$6.127 \times 10^{-5}$	0.8614	$1.930 \times 10^5$	$4.980 \times 10^{-4}$	1	$1.078 \times 10^5$
4.5	0	$6.463 \times 10^{-5}$	0.7950	$4.998 \times 10^4$	$4.839 \times 10^{-4}$	0.8000	$1.034 \times 10^5$
	100	$6.316 \times 10^{-5}$	0.8000	$1.848 \times 10^5$	$4.247 \times 10^{-4}$	0.8000	$1.200 \times 10^5$
8.5	0	$6.069 \times 10^{-5}$	0.7857	$3.550 \times 10^4$	$4.537 \times 10^{-4}$	0.7986	$6.067 \times 10^4$
	100	$5.776 \times 10^{-5}$	0.8155	$9.142 \times 10^4$	$5.519 \times 10^{-6}$	0.9023	$6.419 \times 10^4$
10.5	0	$8.465 \times 10^{-5}$	0.8000	$2.018 \times 10^4$	$1.680 \times 10^{-3}$	0.8012	$2.320 \times 10^4$
	100	$6.537 \times 10^{-5}$	0.8221	$6.685 \times 10^4$	$9.575 \times 10^{-4}$	0.8000	$4.568 \times 10^4$

Here,  $Z_{\text{CPE}}$  is the impedance of the constant phase element ( $\Omega$  cm<sup>2</sup>);  $\omega$  is the angular frequency of the AC voltage ( $\text{rad s}^{-1}$ );  $Y_0$  is the magnitude of admittance of the CPE ( $\Omega^{-1} \text{ cm}^{-2} \text{ S}^n$ ); and  $n$  is a dimensionless number, the CPE is considered ideal capacitance when  $n = 1$ . Pang et al. (1990) provides a detailed theoretical explanation for this EEC, and Constantin and Chirita (2013) even used it to model the electrochemical corrosion of pyrite in acidic media. The impedance parameters obtained by fitting the EIS data to the equivalent circuit are listed in Table 2.

Comparing the galena electrochemical parameters in 0 mg/L and 100 mg/L HA solutions, we can see that whether the electrolyte was acidic (pH 2.5 or 4.5) or alkaline (pH 8.5 or 10.5), increasing the concentration of HA led to an increase in the  $R_t$  value and a decrease in the double-layer capacitance ( $CPE_{\text{e}}, Y_0$ ). The increase in the  $R_t$  value indicates that ions change after transport through the double-charge layer. The decrease in  $CPE_{\text{e}}, Y_0$  was attributed to the decrease in the local dielectric constant of the electrical double layer. The lower capacitance and the greater resistance of the double layer indicate that increasing the concentration of HA weakened the galena electrochemical oxidation. Furthermore, increasing the concentration of HA caused greater passive film resistance ( $R_f$ ) and smaller passive capacitance ( $CPE_{\text{f}}, Y_0$ ). The smaller capacitance and the greater resistance of the film on the electrode surface indicate a more dense passive film. All of these results agree with the polarization results.

When the electrolyte did not contain HA in the double-charge layer, increased electrolyte acidity (pH decrease from 4.5 to 2.5) or alkalinity (pH increase from 8.5 to 10.5) resulted in smaller charge transfer resistance ( $R_t$ ) and greater double-layer capacitance ( $CPE_{\text{e}}, Y_0$ ). This result indicates that ions can be easily transported through the double-charge layer. Furthermore, in the passive film, increased electrolyte acidity or alkalinity also caused smaller passive resistance ( $R_f$ ) and greater passive film capacitance ( $CPE_{\text{f}}, Y_0$ ), suggesting a looser passive film. All of these results agree with the polarization results that higher acidity/alkalinity accelerates galena electrochemical corrosion.

When the electrolyte contained 100 mg/L HA in the double-charge layer, in contrast to the electrolyte without HA, the decreased electrolyte acidity (pH increase from 2.5 to 4.5) or increased alkalinity (pH increase from 8.5 to 10.5) resulted in smaller charge transfer resistance  $R_t$  and larger double-layer capacitance ( $CPE_{\text{e}}, Y_0$ ). These results indicate that ions can be easily transported through the double-charge layer to the passive film, leading to smaller passive resistance  $R_f$  and larger passive film capacitance ( $CPE_{\text{f}}, Y_0$ ), suggesting a looser passive film. The results reveal that stronger alkalinity increases galena corrosion when the electrolyte contains HA, which is consistent with the results of the solution analysis.

### 3.3. Surface characterization measurements

In order to analyze the interaction between Pb (II) and HA particles, FTIR spectra of HA-Pb (II) were obtained and compared with the FTIR spectrum of HA. Fig. 4 shows the FTIR spectra of the massive galena electrodes after weathering in pH 4.5 and pH 8.5 solutions with different concentrations of HA. When no HA was present, the FTIR

spectroscopic proved the black solutions no HA group band. It is worth noting that the bands between 2000 cm<sup>-1</sup> and 2500 cm<sup>-1</sup> are attributed to CO<sub>2</sub>, as the test was performed in air. To HA, the FTIR spectroscopic data exhibited three distinct peaks, namely, carboxyl at 1600 cm<sup>-1</sup> (–COO<sup>-</sup> stretch and possibly aromatic C=C stretch), phenol at 1385 cm<sup>-1</sup> and 1035 cm<sup>-1</sup>, and saturated group (–CH<sub>2</sub> symmetric stretching and asymmetric stretching region 2840–2940 cm<sup>-1</sup>) (Baes and Bloom, 1989; Niemeyer et al., 1992).

At pH 4.5 and in the presence of HA, the galena sample weathered for 10 weeks exhibited three pronounced spectral features: (1) a new vibration peak at 1710 cm<sup>-1</sup>, which was attributed to –C=O stretch in COOH (Swift, 1996), the cause was –COO<sup>-</sup> and H<sup>+</sup> band to –COOH group; (2) compare with another appeared simultaneously wave-number band 1385 cm<sup>-1</sup>, the peak at 1035 cm<sup>-1</sup> almost disappeared for HA and Pb (II) ions complex; (3) the peak at 1600 cm<sup>-1</sup> for HA carboxyl group shifted to an unresolved band at about 1458 cm<sup>-1</sup>, which further confirmed the formation of HA-Pb(II) complexes. The  $\Delta$  parameter value was 142 cm<sup>-1</sup>, indicating the formation of bridging complexes (Fig. 1a) (Nakamoto, 1989). Jerzykiewicz (2004) obtained a similar result in the study of HA interactions with Pb (II) ions.

At pH 8.5 and in the presence of HA, the galena sample weathered for 10 weeks exhibited the following spectral features: (1) when no additional Pb<sup>2+</sup> ions were added, whether in 10 mg/L or in 100 mg/L HA soak solution, no HA or HA-Pb (II) occur. This is because Pb<sup>2+</sup> ions from PbS were oxidized and transformed into Pb(OH)<sub>2</sub>, which further proved the oxidation process of galena and produce more S<sup>0</sup>. S<sup>0</sup> film covered on galena surface and thus resisted HA adsorbed on galena surface; (2) when 1 mg/L Pb<sup>2+</sup> ions were added to the HA solution, on one hand, the added Pb<sup>2+</sup> ions inhibit galena oxidation, not enough S film covered the galena surface. On the other hand, the added Pb<sup>2+</sup> ions can interaction with HA, and causing a similar FTIR result as that in pH 4.5.

Fig. 5 shows the Raman spectra of pristine and weathered galena samples. In the pristine samples, Raman active phonon for the cubic structure of galena was not observed. After the samples were weathered for 10 weeks, the galena sample weathered at pH 4.5 with 100 mg/L HA exhibited three Raman peaks at 152 cm<sup>-1</sup>, 216 cm<sup>-1</sup> and 437 cm<sup>-1</sup> (Fig. 5(a)). The Raman peaks at 152 cm<sup>-1</sup> and 216 cm<sup>-1</sup> were assigned to the vibration mode of the S-S bond of elemental sulfur (Li et al., 1993). The Raman peak at 437 cm<sup>-1</sup> is characteristic of a structure for the anglesite species (Shapter et al., 2000; Batonneau et al., 2000). At pH 4.5 and without HA, the galena spectrum exhibited broad Raman peaks at 435–447 cm<sup>-1</sup>, 610 cm<sup>-1</sup>, 975 cm<sup>-1</sup> and 1051 cm<sup>-1</sup> (Fig. 5(b)). All peaks were assigned to the anglesite species (Shapter et al., 2000; Batonneau et al., 2000). Comparing the galena spectra in acidic conditions with and without HA, galena is oxidized to sulfur during the initial step and ultimately transformed into anglesite during the galena weathering process. HA can easily adsorb on the galena surface and inhibit its transformation into sulfur, as well as inhibit galena weathering. These results are in good agreement with the above electrochemical results. When the solution pH was 8.5 and contained 100 mg/L HA, the galena Raman spectrum exhibited two typical Raman peaks of anglesite at 435–439 cm<sup>-1</sup> and 973 cm<sup>-1</sup> (Fig. 5(c)).

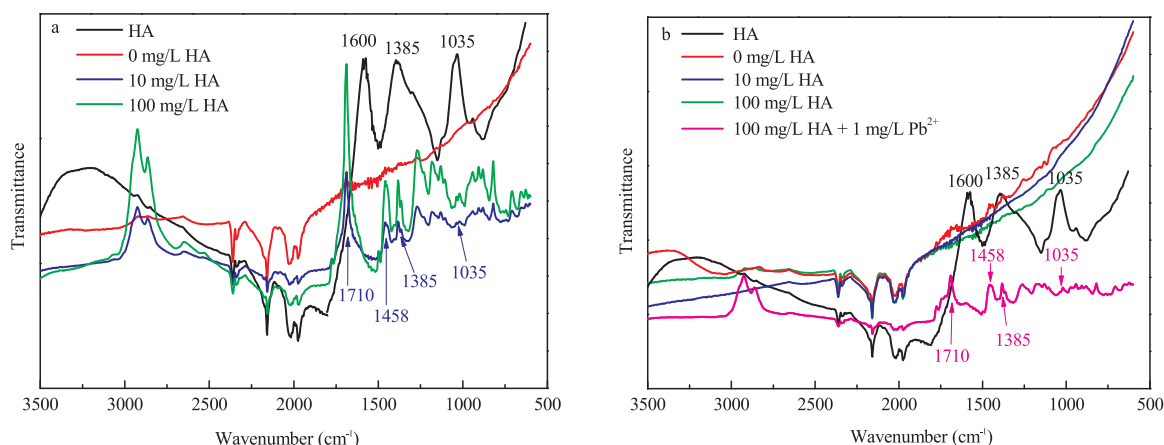


Fig. 4. FTIR spectra of galena after weathering 10-weeks in (a) pH 4.5 and (b) pH 8.5 solutions with different concentrations of HA.

However, when the solution did not contain HA, three different peaks were observed (Fig. 5(d)). The peak at  $216\text{ cm}^{-1}$  was assigned to sulfur as mentioned above, and the peak at  $291\text{ cm}^{-1}$  resulted from the orthorhombic  $\beta$ -PbO species. The third peak at  $346\text{ cm}^{-1}$  resulted from tetragonal  $\alpha$ -PbO species (Wiechert et al., 2005). The galena spectra under alkaline conditions with and without HA revealed two different weathering mechanisms. At the initial step, galena is oxidized to sulfur and  $\text{Pb}^{2+}$ . When the solution contained HA, sulfur can be further reduced to sulphate ions because of the HA ionization effect. Then,  $\text{SO}_4^{2-}$  and  $\text{Pb}^{2+}$  combine to form  $\text{PbSO}_4$ . However, in the solution without HA, sulfur cannot be reduced, and  $\text{Pb}^{2+}$  combines with  $\text{OH}^-$  to form  $\text{Pb}(\text{OH})_2$ . During the Raman spectroscopic measurements, it seemed that

$\text{Pb}(\text{OH})_2$  might be in the form of  $\text{PbO}\cdot\text{H}_2\text{O}$ , and its Raman peak manifests as a waveform of PbO. All of these results are consistent with the electrochemical results. As stated above, Fig. S3 shows galena the weathering mechanism with and without HA via electrochemical and surface analyses.

### 3.4. Environmental risk assessment based on in situ electrochemical and surface investigation

In nature, galena and HA usually occur together. Galena weathering releases  $\text{Pb}^{2+}$  and causing heavy metal ions pollution. HA can bind metal ions; therefore, the environmental issues from galena (include

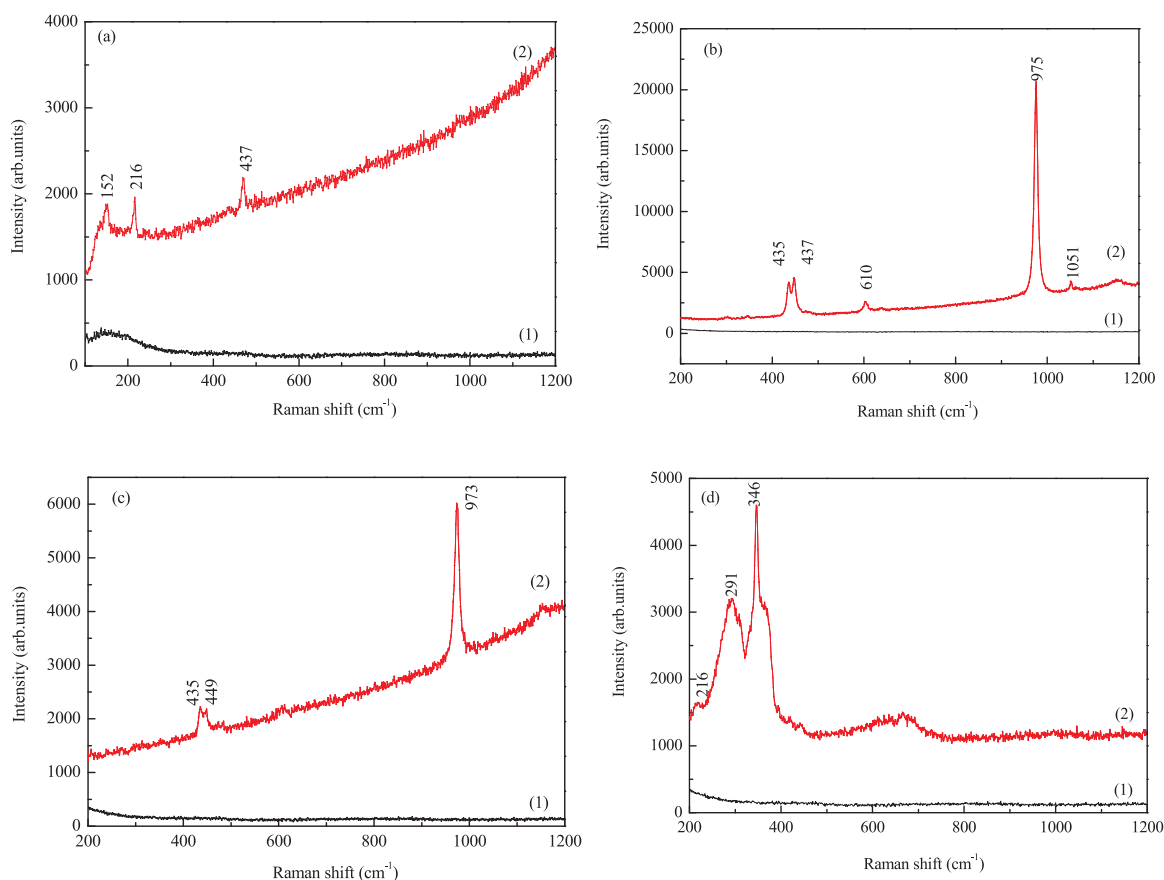


Fig. 5. Raman spectra of the massive galena electrodes after weathering in different acid/alkaline solutions with/without HA. Where: (a) pH 4.5 with 100 mg/L HA, (b) pH 4.5 without HA, (c) pH 8.5 with 100 mg/L HA and (d) pH 8.5 without HA for: (1) pristine surface; (2) 10-weeks leached surfaces.

other metal sulfide minerals) weathering must be affected by the presence of HA. At present, acid mine drainage is one of the serious environmental issues surrounding metal sulfide mineral mining, and the accumulated  $H^+$  ions will accelerate galena weathering and aggravate heavy metal ion pollution. The presence of HA can inhibit galena weathering and, thus, mitigate environmental pollution in mining districts and the surrounding areas.

Generally speaking, higher acidity or higher alkalinity promotes galena corrosion and results in more serious environmental issues during galena weathering process. In the future, HA, especially after physical-chemical modification may be a good adsorbent for toxic heavy metals. Depending on the corrosion current density  $i_{corr}$ , the galena weathering rate could be calculated according to the Faraday equation (Eq. 9) (Gunawarman et al., 2016):

$$v = \frac{Mi_{corr}}{nF} \quad (9)$$

where  $v$  is the weathering rate in  $g\ m^{-2}\ h^{-1}$ ,  $i_{corr}$  is corrosion current density in  $\mu A\ cm^{-2}$ ,  $M$  is the atomic weight in  $g\ mol^{-1}$ ,  $n$  is the valence state, and  $F$  is the Faraday constant ( $96,487\ C\ mol^{-1}$ ). For example (Table 1), for a solution pH of 2.5 without HA, galena  $i_{corr}$  was  $0.398\ \mu A\ cm^{-2}$ . Thus, the galena weathering rate was  $0.015\ g\ m^{-2}\ h^{-1}$ . Therefore, when the galena area is  $1\ m^2$ , it will release  $134.7\ g\ Pb^{2+}$  to the surrounding environment per year. When the solution contained  $1000\ mg/L$  HA, galena  $i_{corr}$  decreased to  $0.083\ \mu A\ cm^{-2}$ , and the galena weathering rate was  $0.0032\ g\ m^{-2}\ h^{-1}$ . Based on the calculations, the amount of  $Pb^{2+}$  released will decrease to  $28.09\ g$  per year, under the same conditions. This method could be used to assess how much  $Pb^{2+}$  ions will be released over a certain time and how much HA is required to mitigate the release of  $Pb^{2+}$ . In addition, this method can be applied to soil, if the content of galena in the soil is known.

#### 4. Conclusions

Galena weathering was studied under acidic and alkaline conditions depending on electrochemical method combined with spectroscopic techniques, with and without HA. The polarization curve and EIS revealed that HA inhibits galena weathering for causing greater polarization resistance and passive film resistance. HA doesn't change galena weathering mechanism under acidic conditions, and changes the galena weathering mechanism via ionization effect under alkali conditions. Galena weathering releases  $134.7\ g/m^2/y\ Pb^{2+}$  to the surrounding environment when pH is 2.5, and the amount released decreases to  $28.09\ g$  in the presence of  $1000\ mg/L$  HA. The FTIR spectra and Raman spectra revealed that HA and Pb (II) ions form bridging complexes under acidic conditions. Under alkali conditions without HA, galena weathering ultimate products are  $S^0$  and  $PbO\cdot H_2O$ , and the ultimate product is anglesite the presence of HA.

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#### Appendix A. Supplementary material

Supplementary data associated with this article can be found in the online version at <http://dx.doi.org/10.1016/j.ecoenv.2018.04.030>.

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