



Pressure solution of electrically conductive minerals in shallow crust-galvanic processes: A case study from pyrite under differential stress



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ABSTRACT

The pressure dissolution behaviour of pyrite was investigated via its polarisation curve and using electrochemical impedance spectroscopy (EIS) in a FeCl_3 solution under differential stress. The results showed that the pyrite pressure dissolution process is a galvanic corrosion process and that there is a negative linear relationship between the pyrite potential difference and the stress action. The EIS experiments confirm that the pyrite was in a passive state in a $0.0010 \text{ mol L}^{-1}$ FeCl_3 solution and that a thin surface layer of $\text{Fe}_{1-y}\text{S}_2$ was present. In a 0.010 mol L^{-1} FeCl_3 solution, the pyrite was in a trans-passive state, in which the aforementioned passive layer became porous. In a 0.10 mol L^{-1} FeCl_3 solution, the pyrite was in an active state, the surface layer dissolved completely, and a lattice layer of S_2^0 was created instead of a passive layer of S^0 . Under the present stress conditions, the stress action did not change the pyrite electrochemical dissolution mechanism; however, the conditions decreased the charge transfer resistance and passive resistance and increased the species diffusion capacitance.

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1. Introduction

In geology, pressure solution is defined as the process by which grains dissolve at intergranular and intercrystalline contacts (Tada and Siever, 1989). This phenomenon is a long-standing problem that has engendered continuing arguments with respect to its mechanism, thermodynamic basis, and geological significance. In fact, pressure solution has been used to explain many phenomena in geology, such as stylolitic structure (Guzzetta, 1984; Koehn et al., 2006; Nenna and Aydin, 2011), tectonic accretion (Fueten et al., 2002; Beaumont et al., 2009) and element migration and element activation (Seyfried and Janecky, 1985; Meer et al., 2002; Loring et al., 2004; Birgersson and Karnland, 2009).

In the past few decades, pressure solution has received renewed attention for its important role in the diagenesis of sedimentary rocks and its relationship to rock deformation. Pressure solution has been observed as a fluid migration mechanism (Angevine and Turcotte, 1983), a hydrocarbon accumulation mechanism (Carozzi and Bergen, 1987), and a deformation mechanism under diagenetic conditions (Marshak and Engelder, 1985). Liu and Wu (1994) showed that pressure solution was usually observed as a mechanical–chemical dissolution process. According to the authors' previous research, the above mechanism can sufficiently explain the pressure solution of non-conductive minerals, but

conductive minerals, such as metallic sulfides, some oxides and natural metallic minerals, may have a different pressure solution mechanism (Li et al., 1998).

Pyrite, the most abundant sulfide in the Earth's crust (Peters and Majima, 1968), has motivated most of the research in the literature because of its potential use as a component in solar energy collectors (Gerischer, 1975; Chen et al., 1991), depolariser anodes for H_2 production (Lalvani and Shami, 1986), and cathodes in high-energy-density batteries (Vincent et al., 1984). The electrochemical dissolution of pyrite plays an important role in the fields of environmental protection and resource development through phenomena such as the formation of acidic mine drainage (Singer and Stumm, 1970), the biogeochemical cycles of S, Fe and other metals (Nordstrom, 1982), the microbiology of Fe- and S-oxidising bacteria (Sasaki et al., 1994), the mineral processing of sulfides (Murr, 1980), and coal cleaning (Silverman et al., 1961). As a result, there is an extensive literature investigating the dissolution of pyrite that has been published in the past ten years alone (Cabral and Ignatiadis, 2001; Cruz et al., 2001; Chernyshova, 2003, 2004; Lehner et al., 2007; Savage et al., 2008; Liu et al., 2008, 2009b; Heidel and Tichomirowa, 2010).

The authors are not aware of any reports associated with the electrochemical behaviour of pyrite (FeS_2) under stress. However, sulfide minerals under human exploitation or endemic geochemical processes constantly experience stress. For instance, in the Earth's crust, differential stress derived not only from gravitational force, caused by the overlying rocks but also from tectonic, thermal

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and other dynamic forces. These differential stresses control the deformation and the fabric variation of rocks (Hast, 1969, 1974). In addition, in the hydrometallurgy field, mechanical activation is often viewed as an effective measure to enhance the leaching processes, needless to say, mechanical action leads to minerals storing residual stress, that is, minerals store the strain energy, which can be converted into electrochemical energy. The strain energy produced by stress can be converted into electrochemical energy, which subsequently influences the electrode potential and electrochemical behaviour of these sulfide minerals, and eventually results in leaching rate enhancement. Ferric chloride (FeCl_3) is not only a common electrolyte, it is also the most-used medium for sulfide leaching (Dutrizac, 1992), a common Fe(III) compound in the Earth's crust (Bailey et al., 2002) and a common component of mine waters (Bowell and Bruce, 1995; Silva et al., 2009); consequently, it has a strong influence on the electrochemical behaviour of metal sulfides.

In view of the above discussion, the aim of this work is to study the electrochemical response of pyrite in a FeCl_3 solution under differential stress to understand the dissolution kinetics and electrochemical mechanisms. This work will provide the experimental basis for a pressure solution mechanism of pyrite in the shallow crust and will provide the mechanism by which geo-stress affects metal and acidic pollution arising from metal sulfide mine drainage.

2. Experimental methods

2.1. Samples, reagents and instrumentation

Natural pyrite from the Yunfu Fe deposit (Guangdong Province, China) was used in this work. The sample was cut and shaped into a $10 \text{ mm} \times 10 \text{ mm} \times 40 \text{ mm}$ block. Within the limits of these techniques, reflected light microscopy and X-ray diffraction analyses indicated that the sample was pure and homogeneous. The electron microprobe analysis indicated that the Fe and S contents (wt%) were 46.89% and 52.96%, respectively. Before the experiments, the electrode surface was polished with metallographic abrasive paper, cleaned with filter paper and then rinsed with acetone. After each experiment, the pyrite electrode was re-polished and re-used.

Analytical grade FeCl_3 and double distilled water were used to prepare a series of FeCl_3 solutions of various concentrations. During all experiments, a 60 mL electrochemical cell made of rigid polytetrafluoroethylene was filled with FeCl_3 solution, without purging the dissolved O_2 .

All mineral stress experiments were performed on an electronic universal testing machine controlled by a computer. Two resistance strain gauges were installed on the pyrite sample surface,

along the axial and lateral directions. Then, depending on the dynamic and static strain testing systems, axial stress, axial strain and lateral strain data could be obtained. To study the influence of stress, rapid and continuously varying axial stresses (0 Pa ; $2.0 \times 10^5 \text{ Pa}$; $4.0 \times 10^5 \text{ Pa}$; $6.0 \times 10^5 \text{ Pa}$) were applied in the electrode potential experiment. The respective polarisation curves and electrical impedance spectroscopy data were collected.

The experiments were conducted using a stationary electrode at an ambient temperature of $25 \pm 1 \text{ }^\circ\text{C}$.

2.2. Electrode potential measurements under various differential stresses

An Agilent 34410A high-precision digital multimeter connected to a computer was used to measure the electrode potential. A standard two-electrode system with an exposed pyrite area of 4 cm^2 was constructed from a pyrite electrode and a saturated calomel electrode (SCE) connected through a Luggin capillary salt bridge. All potentials recorded in this study were relative to the SCE. The experimental configuration is shown schematically in Fig. 1.

2.3. Polarisation curves and electrical impedance spectroscopy measurements

Polarisation curve measurements were performed using a computer-controlled electrochemical measurement system on a conventional three-electrode electrolytic cell that used Pt as an auxiliary electrode and pyrite as the working electrode. A saturated calomel electrode (SCE) was used as the reference electrode for all electrochemical tests, and all potentials quoted in this study are with respect to the SCE (0.242 V vs. SHE), unless otherwise indicated. The reference electrode was connected to a Luggin capillary to minimise the IR drop. The configuration of the working, auxiliary and reference electrodes were maintained to ensure that each experiment had the same spatial orientation.

The polarisation scan was performed from -250 mV relative to the open circuit potential (E_{OC}) to $+250 \text{ mV}$ relative to the E_{OC} at a rate of 1 mV/s . Data were recorded after a 60 min immersion in the working solution. The experimental control and data analyses were performed using the PowerCorr software provided by Princeton Applied Research.

The EIS measurements were performed at the open circuit potential in the $1.0 \times 10^4 \text{ Hz}$ to 0.1 Hz frequency range using a Parstat-2263 interfaced with a PC. The amplitude of the sinusoidal voltage was 5 mV rms . Data were recorded after a 60 min immersion in the working solution. The equivalent circuit model used for fitting the AC impedance was evaluated with the ZSimpWin (Version 3.10) software from PAR.

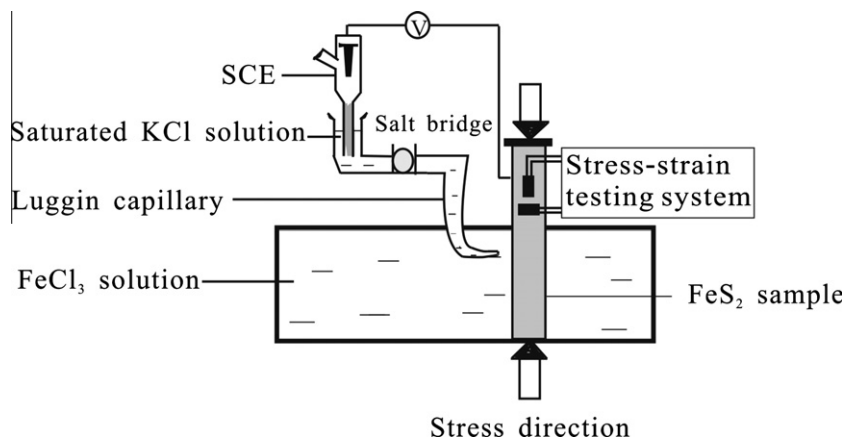


Fig. 1. Schematic diagram showing the stress-potential experimental arrangement.

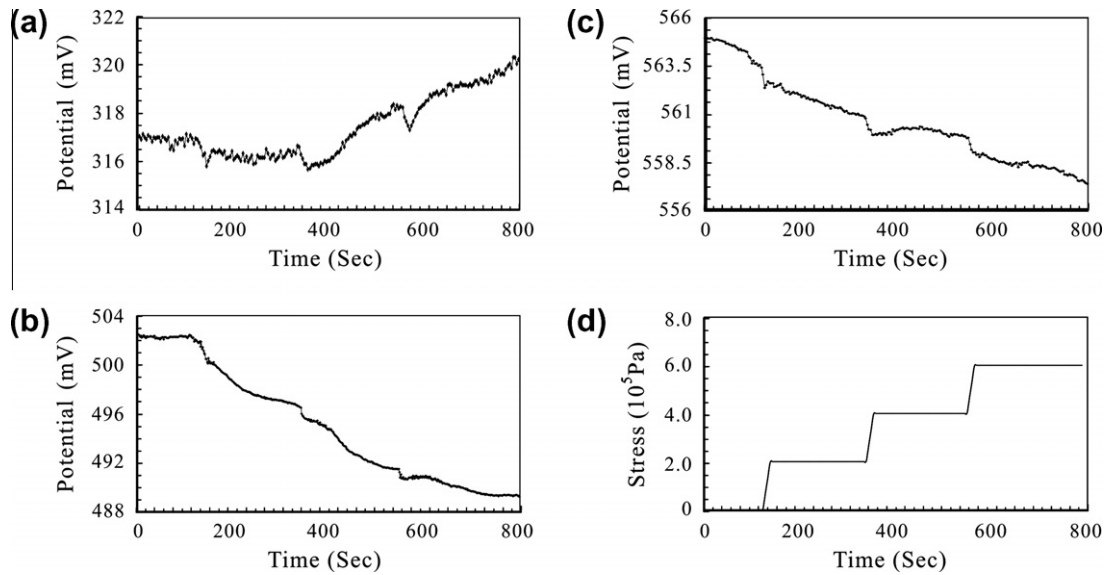


Fig. 2. Variations in pyrite electrode potential under rapid, continuously varying axial stresses (d) in different concentrations FeCl_3 solution. (a) $0.0010 \text{ mol L}^{-1}$, (b) 0.010 mol L^{-1} and (c) 0.10 mol L^{-1} .

3. Results and discussion

3.1. Stress, strain and electrode potential

Fig. 2 shows the electrode potentials of pyrite in FeCl_3 solutions ($0.0010 \text{ mol L}^{-1}$; 0.010 mol L^{-1} ; 0.10 mol L^{-1}) under rapid and continuous varying differential stresses. From Fig. 2, it can be seen that the electrode potential of pyrite differed for different FeCl_3 solution concentrations under similar differential stresses, but the electrode potential differences remained the same. The combined stress and strain conditions in the experiment allowed for derivation of a stress–strain–potential chart (Fig. 3). From Fig. 3, it can be seen that the axial stress and the axial strain had a positive linear relationship, as did the axial stress and the lateral strain. Both relationships obey Hooke's law. These results indicate that the axial stresses on the pyrite were elastic and that the pyrite experienced elastic strain. From the figure, it can also be seen that the axial stress and the pyrite potential were linked by a negative linear relationship.

The pyrite potential is defined as $\varphi_H^{\text{FeS}_2}$. When the pyrite is under elastic stress conditions, mechanical interactions do not affect the electrolyte solution, and the chemical potentials of the reducible ions in the solution do not change, although the chemical potential of the pyrite changes. It can be determined that the electrode potential difference for stress is $\Delta\varphi_H^{\text{FeS}_2} = -\frac{V\Delta p}{ZF}$ (Liu et al., 2009a), where V is pyrite molar volume, Z is electron transfer number and F is Faraday constant. This equation shows that there is a negative linear relationship between the pyrite potential difference and the stress action. The experimental results shown in Fig. 2 are consistent with this electrochemical theory.

3.2. Study of polarisation curves

Fig. 4 shows the Tafel plots for pyrite in naturally aerated FeCl_3 solutions ($0.0010 \text{ mol L}^{-1}$; 0.010 mol L^{-1} ; 0.10 mol L^{-1}). From Fig. 4, it can be seen that increasing stress causes the polarisation curve to shift in the positive direction along the horizontal and longitudinal axes with increasing FeCl_3 concentrations. The corrosion current density, the Tafel slope, the transfer coefficient and the number of electrons transferred were calculated according to electrochemical theory (Bard and Faulkner, 2001) using the PAR software.

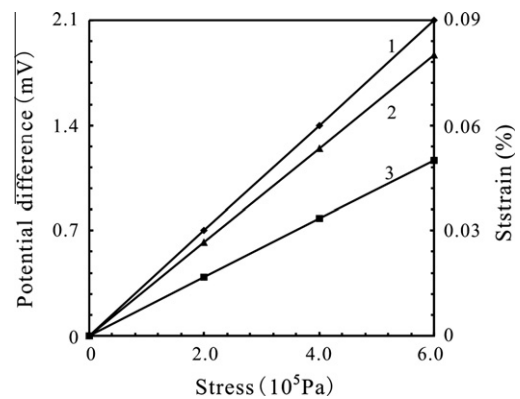


Fig. 3. Chart of pyrite potential difference–stress–strain. (1) axial stress–pyrite potential difference, (2) axial stress–axial strain and (3) axial stress–lateral strain.

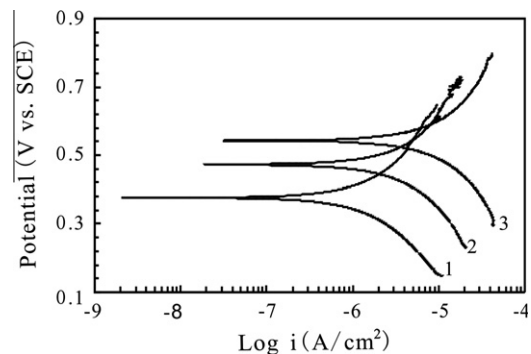


Fig. 4. Tafel plots for pyrite electrode under different concentrations FeCl_3 solutions. (1) $0.0010 \text{ mol L}^{-1}$, (2) 0.010 mol L^{-1} , (3) 0.10 mol L^{-1} .

$$I = i_{\text{corr}} \left[\exp \frac{2.303(E - E_{\text{corr}})}{b_a} - \exp \frac{-2.303(E - E_{\text{corr}})}{b_c} \right] \quad (1)$$

$$b_a = \frac{2.303RT}{\beta nF} \quad (2)$$

$$b_c = \frac{2.303RT}{\alpha nF} \quad (3)$$

$$\alpha = 1 - \beta \quad (4)$$

where E_{corr} is corrosion potential; i_{corr} is corrosion current density; b_a and b_c are Tafel slope of anode and cathode, respectively; α and β are transfer coefficient of anode and cathode, respectively; n is number of electron transfer; T is absolute temperature; R is universal gas constant; and F is Faraday constant.

Furthermore, the polarisation resistance values, R_p , can be calculated from the Stern-Geary equation (Gonzalez-Rodriguez et al., 2009):

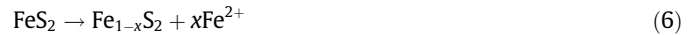
$$I_{\text{corr}} = \frac{b_a b_c}{2.3(b_a + b_c)} \cdot \frac{1}{R_p} \quad (5)$$

The linear polarisation experimental results are listed in Table 1. The pyrite electrode corrosion current density increased from 5.857 to 190.7 $\mu\text{A cm}^{-2}$ as the FeCl_3 concentration increased from 0.0010 to 0.10 mol L^{-1} , the corrosion potential became more positive, increasing from 363.161 to 541.407 mV, and the polarisation resistance values decreased from 10.188 to 1.770 Ωcm^2 . These results show that increased FeCl_3 concentrations promote the pyrite electrochemical interaction and that they induce more significant galvanic corrosion. It should also be noted that the transfer coefficient and the number of electrons transferred, which are indicative of the electrode reaction mechanism (Liu, 2002), changed noticeably at these three different concentrations. These results suggest that there are different electrochemical interaction mechanisms under these conditions. The probable direct cause of this phenomenon is the ability of Cl^- ions to form complexes and compounds with the available Fe ions in the solution. Additionally, the high mobility of the Cl^- ions permits them to reach and coat the electrode surface, halting the formation of oxide and hydroxide films. This action prevents or postpones the neutralisation of the surface (Pavlica, 1974). To further investigate the anode electrochemical mechanism, impedance studies were performed at open circuit potentials for each FeCl_3 solution concentration.

3.3. Electrical impedance spectroscopy measurements

Fig. 5a and b shows the Bode and Nyquist plots for the pyrite electrode in 0.0010 mol L^{-1} FeCl_3 solution under different elastic axial stresses. The results show that they have similar Bode and Nyquist profiles. This similarity shows that the pyrite electrode has the same electrochemical interaction mechanism under these differing axial stresses (0; 2.0×10^5 Pa; 4.0×10^5 Pa; 6.0×10^5 Pa). As the stress increases, the total impedance magnitude of the system decreases, and the phase angle decreases for high frequencies and increases for the low frequencies. An examination of the Bode plots in Fig. 5a reveals there are two time constants corresponding to two faradaic relaxation processes. The first time constant, which is observed at high frequencies, is a capacitive loop that is related to the charge transfer resistance from the electrochemical dissolution of the electrode. The second time constant, which occurs at the middle-high frequencies, is another a capacitive loop that is related to the combination of the pseudo-capacitive impedance of the passive layer and a resistance.

Consequently, the equivalent circuit shown in Fig. 5c was employed to fit the experimental impedance data. In the equivalent circuit, R_s represents resistances from the electrolyte and other ohmic sources in their respective electrochemical circuits, CPE_t is the constant phase element that corresponds to the double-layer capacitance, R_t is the charge transfer resistance, and the CPE_p/R_p pair represents the capacitive and resistive behaviours of the passive film. Under these conditions, the film is believed to be composed of $\text{Fe}_{1-y}\text{S}_2$ and created by the following reaction (Karthé et al., 1993).



The impedance of the CPE is given by Macdonald (1985):

$$Z_{\text{CPE}} = \frac{1}{Y_0(j\omega)^n} \quad (7)$$

Here, Z_{CPE} is the impedance of the constant phase element (Ωcm^2), ω is the angular frequency of the AC voltage (rad s^{-1}), Y_0 is the magnitude of admittance of the CPE ($\Omega^{-1} \text{cm}^{-2} \text{S}^{-n}$) and n is the exponential term. The values associated with the different elements in the equivalent circuit are shown in Table 2. It can be observed that the space charge transform resistance R_t decreased from 4942 to 2143 Ωcm^2 and that the passive film resistance R_p decreased from 5.011E4 to 1.198E4 Ωcm^2 when the stress increased from 0 to 6.0×10^5 Pa. The Y_0 value of the CPE_t increased from 6.758E-7 to 9.167E-5, and Y_0 of the CPE_p increased from 7.635E-4 to 8.838E-4. In the cases above, the two Y_0 values refer to the double-layer and passive film capacitances, respectively. These capacitances are related to the pyrite dissolution and passive film depletion processes, respectively. A higher Y_0 value indicates an increased transport of ions through the double-charge layer and the passive layer. The high capacitance and low resistance of the films on the specimens indicate the relatively poor robustness of the passive film. These results reveal that greater stresses enhance pyrite pressure dissolution and passive film depletion.

As the concentration of the FeCl_3 solution increases, the pyrite open circuit potential increases, and the existing passive layer is oxidised and becomes increasingly porous to the strong penetrative nature of the Cl^- ions. Fig. 6 shows the Bode plot diagram and a typical Nyquist diagram for pyrite in the presence of a 0.010 mol L^{-1} FeCl_3 solution under different elastic axial stresses. The Bode and Nyquist plots at this concentration have similar profiles to the 0.0010 mol L^{-1} result for different axial stresses, revealing the same corrosion pattern. Furthermore, Fig. 6a and b shows two obvious time constants, as indicated by two distorted capacitive loops. Fig. 6c shows the equivalent circuit employed to fit the experimental data, where R_s describes the electrolytic and other ohmic resistances in the respective electrochemical circuits, and the CPE_t/R_t pair represents the charge transform capacitive and resistive behaviours in the space charge region, respectively. The pair CPE_1/R_1 refers to sections of the passive layer with the maximum thickness and the C_p/R_p pair refer to sections of the passive layer with the minimum thickness (Mansfeld and Kendig, 1988). The constant phase element CPE is used to account for the capacitance, which probably arises from electrode surface defects. The values of the different elements in the equivalent circuit are shown in Table 3. The results in Table 3 are for the case when the elastic

Table 1
Electrochemical kinetic parameters of the FeS_2 samples from Tafel tests.

FeCl_3 (mol L^{-1})	E_{corr} (mV)	i_{corr} ($\mu\text{A cm}^{-2}$)	b_a (mV)	b_c (mV)	α	β	n	R_p (Ωcm^2)
0.0010	363.161	5.857	236.066	327.861	0.419	0.581	0.0518	10.188
0.010	458.317	51.18	678.283	1612.299	0.296	0.704	0.148	4.056
0.10	541.407	190.7	1286.320	1958.020	0.396	0.604	0.914	1.770

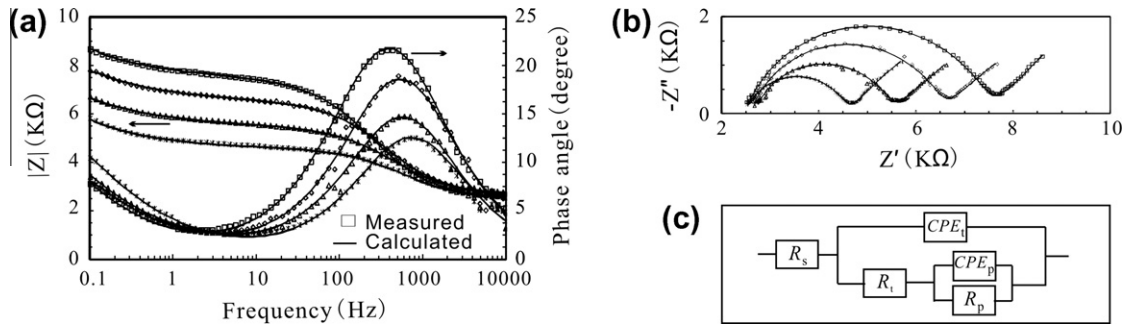


Fig. 5. Bode plots and phase angles (a), Nyquist impedance spectra (b) and equivalent circuit (c) for pyrite under differential stresses in 0.0010 mol L⁻¹ FeCl₃ solutions. (□) 0 Pa, (◇) 2.0 × 10⁵ Pa, (Δ) 4.0 × 10⁵ Pa and (×) 6.0 × 10⁵ Pa.

Table 2
Impedance parameters of FeS₂ in 0.0010 mol L⁻¹ FeCl₃ solutions under differential stresses.

FeCl ₃ (mol L ⁻¹)	Stress (10 ⁵ Pa)	CPE _t , Y ₀ (S cm ⁻² s ⁻ⁿ)	n	R _t (Ω cm ²)	CPE _p , Y ₀ (S cm ⁻² s ⁻ⁿ)	n	R _p (Ω cm ²)
0.0010	0	6.758E-7	0.8107	4942	7.635E-4	0.5026	5.011E4
	2	6.894E-7	0.8017	3885	8.037E-4	0.4873	4.189E4
	4	8.070E-5	0.7867	2821	8.247E-4	0.4652	1.680E4
	6	9.167E-5	0.7716	2143	8.838E-4	0.4613	1.198E4

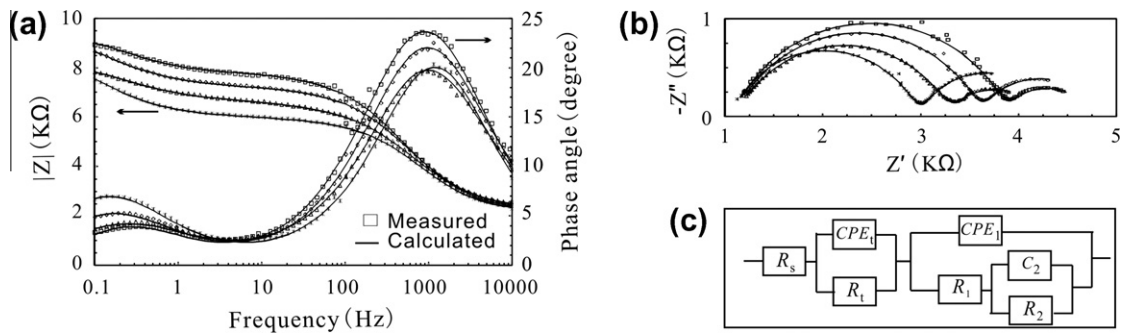


Fig. 6. Bode plots and phase angles (a), Nyquist impedance spectra (b) and equivalent circuit (c) for pyrite under differential stresses in 0.010 mol L⁻¹ FeCl₃ solutions. (□) 0 Pa, (◇) 2.0 × 10⁵ Pa, (Δ) 4.0 × 10⁵ Pa and (×) 6.0 × 10⁵ Pa.

Table 3
Impedance parameters of FeS₂ in 0.010 M FeCl₃ solutions under differential stresses.

FeCl ₃ (mol L ⁻¹)	Stress (10 ⁵ Pa)	CPE _t , Y ₀ (S cm ⁻² s ⁻ⁿ)	n	R _t (Ω cm ²)	CPE ₁ , Y ₀ (S cm ⁻² s ⁻ⁿ)	n	R ₁ (Ω cm ²)	C ₂ (F cm ⁻²)	R ₂ (Ω cm ²)
0.010	0	1.179E-3	0.7568	2692	9.773E-7	0.7541	4.653E4	4.746E-6	3019
	2	1.426E-3	0.6037	2170	9.829E-7	0.7544	3.396E4	1.420E-5	2714
	4	1.627E-3	0.7615	1984	1.002E-6	0.7732	1.322E4	3.640E-5	2254
	6	1.857E-3	0.7776	1791	8.993E-5	0.6910	7.871E3	9.109E-5	1479

stresses are increased from 0 to 6.0 × 10⁵ Pa and are summarised as follows: (1) the charge transfer resistance decreases from 2692 to 1791 Ω cm², and the capacitance parameter increases from 1.179E-3 to 1.857E-3 S cm⁻² s⁻ⁿ; (2) the resistance of sections of the passive layer with the maximum thickness decreases from 4.653E4 to 7.871E3 Ω cm², and the capacitance parameter increases from 9.773E-7 to 8.993E-5 S cm⁻² s⁻ⁿ; and (3) the resistance of sections of the passive layer with the minimum thickness decreases from 3019 to 1479 Ω cm², and the capacitance value increases from 4.746E-6 to 9.109E-5 F cm⁻². These results demonstrate the following trends: (1) the charge transfer resistance decreases and the double layer capacitance parameter increases as the pyrite stress increases, which indicates that stress promotes pyrite dissolution; (2) the resistance of the passive layer in sections with the maximum/minimum thicknesses decreases as

the stress increases, whereas the capacitance value of sections of the passive layer with the maximum/minimum thicknesses increases. These trends reveal that stress enhances passive pitting corrosion.

As the concentration of the FeCl₃ solution increases, open circuit potential and corrosion potential of the pyrite increase. Fig. 7a and b presents the Bode and Nyquist plots, respectively, for the pyrite electrode in the 0.10 mol L⁻¹ FeCl₃ solution under different elastic axial stresses. The similarity of these plots reveals that the same electrochemical mechanism occurs under different values of stress. However, Fig. 7a and b do not exhibit a clear capacitive loop in the low frequency range. It can be concluded that a passive film was not present with a high concentration of Cl⁻ ions. This outcome could be explained by the mechanism proposed by Ahlberg and Broo (1997): in this potential region, the initial oxidation proceeds

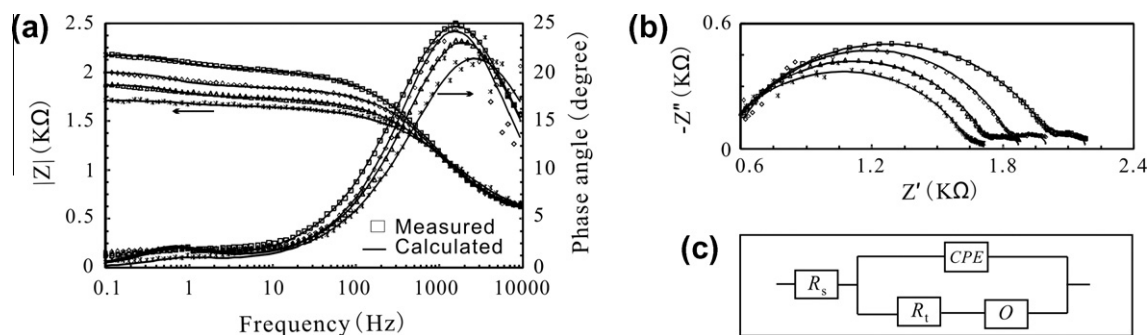


Fig. 7. Bode plots and phase angles (a), Nyquist impedance spectra (b) and equivalent circuit (c) for pyrite under differential stresses in 0.10 mol L⁻¹ FeCl₃ solutions. (□) 0 Pa, (◇) 2.0 × 10⁵ Pa, (Δ) 4.0 × 10⁵ Pa and (×) 6.0 × 10⁵ Pa.

Table 4
Impedance parameters of FeS₂ in 0.10 M FeCl₃ solutions under differential stresses.

FeCl ₃ (mol L ⁻¹)	Stress (10 ⁵ Pa)	CPE, Y ₀ (S cm ⁻² s ⁻ⁿ)	n	R _t (Ω cm ²)	Coth, Y ₀ (S cm ⁻² s ^{-0.5})	B (s ^{-0.5})
0.10	0	8.262E-7	0.7606	1307	4.280E-3	0.8356
	2	8.537E-7	0.8082	1275	4.699E-3	0.7892
	4	1.377E-6	0.7402	1246	6.714E-3	0.7091
	6	1.640E-6	0.7061	1171	7.424E-3	0.6762

without destruction of the lattice. The electrons involved possess energy levels of both Fe and the lattice layer S₂⁰, as indicated by reaction (7). The presence of S₂⁰ instead of S⁰ indicates that elemental S may not be formed, so it is not obvious that a passive product is created.



The equivalent electrical circuit shown in Fig. 7c was used to fit the experimental data, where R_s describes the electrolytic and other ohmic resistances in the respective electrochemical circuits. The time constant in the high-frequency region is the fast charge transfer process between the pyrite and the Fe³⁺ species. R_t is the charge transfer resistance. A constant phase element (CPE) was substituted for the double-layer capacitance connection of the Helmholtz layer and the surface film. The Warburg element O represents the Fe³⁺ species diffusion process, which describes dimensional diffusion through a layer of finite thickness. From an electrical perspective, this representation is equivalent to the impedance of a transmission line of finite length. The different model parameters are shown in Table 4. The following changes were observed when the elastic stress increased from 0 to 6.0 × 10⁵ Pa: the charge transfer resistance decreased from 1307 to 1171 Ω cm²; the capacitance parameter increased from 8.262E-7 to 1.640E-6 S cm⁻² s⁻ⁿ; the capacitance parameter (coth) increased from 4.280E-3 to 7.424E-3 S cm⁻² s⁻ⁿ; and the time constant decreased from 0.8356 to 0.6762 s^{-0.5}. These results show the following: (1) similar to the previous results, stress promotes pyrite pressure dissolution; (2) increasing stress causes the capacitance parameter (coth) to increase and the time constant to decrease, which indicates stress-enhanced dimensional Fe³⁺ diffusion through a layer of finite thickness.

4. Conclusions

The pressure dissolution behaviour of pyrite (FeS₂) in a FeCl₃ solution under differential stress has been studied using polarisation curves and electrical impedance spectroscopy. The following conclusions can be drawn:

- (1) For electrically conductive minerals (such as pyrite) in the shallow crust, when an electrolyte is present, their pressure dissolution process is a galvanic corrosion process.

- (2) There is a negative linear relationship between the pyrite potential difference and the stress action, and the potential is only affected by the characteristics of the mineral. The mathematical expression is the following: $\Delta\phi_{\text{H}}^{\text{FeS}_2} = -\frac{v\Delta P}{ZF}$.
- (3) Different transfer coefficients and numbers of electrons transferred reveal that pyrite has different electrochemical dissolution mechanisms in 0.0010 mol L⁻¹, 0.010 mol L⁻¹ and 0.10 mol L⁻¹ FeCl₃ solutions. Furthermore, increasing the concentration of the FeCl₃ solution enhances pyrite electrochemical dissolution.
- (4) EIS experiments confirmed the different dissolution mechanisms of pyrite. In a 0.0010 mol L⁻¹ FeCl₃ solution, the pyrite is in a passive state, and a thin surface layer of Fe_{1-y}S₂ is present. In a 0.010 mol L⁻¹ FeCl₃ solution, the pyrite is in a trans-passive state, and the aforementioned passive layer becomes porous. In a 0.10 mol L⁻¹ FeCl₃ solution, the pyrite is in an active state. The surface layer is completely dissolved, and a lattice layer S₂⁰ is created instead of a passive layer S⁰. Under the experimental conditions, the stress action did not change the pyrite electrochemical dissolution mechanism. However, in all stress experiments, the stress action decreased the charge transfer resistance and the passive resistance. Additionally, the stress action increased the species diffusion capacitance by significantly enhancing the ionic transport and diffusion processes.

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